

IDENTIFICATION OF TRANSPORT PARAMETERS FOR GALLIUM NITRIDE BASED SEMICONDUCTOR DEVICES

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Abstract. We present a methodology for the identification of transport parameters for Gallium Nitride (GaN) based semiconductor materials and devices. A Monte Carlo (MC) approach has been employed to investigate the electron transport in GaN and AlGa_N, materials that are very important in device applications of high-power, high-frequency electronics. Our model is validated against measured data and compared to published simulation results. It enables to understand effects taking place in this material system, and it provides inputs for macroscopic modeling of electronic devices. Various MC model parameters and simulation results are compared. A special approach on the piezoelectric scattering mechanism taking care of the hexagonal crystal structure is used.

1. Motivation

Material models which incorporate the basic characteristics of the underlying physics in a given semiconductor material are the core of device modeling. While for Silicon such models are well established, models for GaN and GaN-related materials are a hot topic of present research activities. This material system recently became of interest for applications in optical, high-power, and high-frequency electronics. However, GaN still poses many technological as well as modeling challenges. Progress in assessing the entire material information is impeded experimentally by varying material quality and theoretically by the lack of detailed knowledge of relevant parameters. Examples are deformation potentials, piezo-parameters, etc.

2. Considerations on the Monte Carlo Approach

The MC method is a powerful technique to establish a consistent link between theory and experiments. It helps to gain understanding of the transport properties and it provides macroscopic parameters which are necessary for the description of electronic devices. We employ a single-particle MC technique to investigate stationary electron transport in GaN. Our model includes the three lowest valleys of the conduction band (Γ_1 , U, Γ_3). Several stochastic mechanisms such as acoustic phonon, polar optical phonon, inter-valley phonon, ionized impurity scattering, and piezoelectric scattering are considered and their impact is assessed. The particular advantage of the MC method is that it provides a transport formulation on microscopic level, limited only by the extent to which the underlying physics of the system is included. Since the GaN material system is yet not so well explored, several important input parameters are still missing or just inaccurately known. We assess in an iterative approach the influence of the input parameters and their interdependencies in order to get a set of parameters which gives agreement with experimental data available for different physical conditions (doping, temperature, field, etc.). Such a calibrated set of models and model parameters delivers valuable data for low-field mobility, velocity saturation, energy relaxation times, etc. These calibrated models can serve as a basis for the development of analytical models for the simulation of GaN-based electron devices.

An accurate and often sufficient approximation for the conduction band is the parabolic approximation of the three lowest valleys. For values of the wave-vector far from the minima of the conduction band the energy deviates from the simple quadratic expression, and non-parabolicity occurs. The non-parabolic relationship between energy ϵ and wave vector k is given by: $\gamma(\epsilon(k)) = \epsilon (1 + \alpha \epsilon) = \hbar^2 k^2 / 2 m$, with α being the non-parabolicity factor and m the effective electron mass.

In order to establish a rigorous MC simulation, parameters from various publications have been collected and analyzed. Table 1 and Table 2 provide a detailed summary of bulk material parameters for GaN and AlN, respectively, published throughout the past years. These tables include information necessary for analytical band-structure MC simulations, such as energies of lowest conduction bands, effective electron masses, non-parabolicity factors, and

Table 1: Summary of material parameters of wurtzite GaN for Monte Carlo simulation

Bandgap energy			Electron mass			Non-parabolicity			Scattering models					Reference	
Γ_1	U	Γ_3	m_{Γ_1}	m_U	m_{Γ_3}	α_{Γ_1}	α_U	α_{Γ_3}	ADP	$\hbar\omega_{ij}$	$\hbar\omega_{LO}$	ρ	ε_s	ε_∞	Year
[eV]	[eV]	[eV]	[m_0]	[m_0]	[m_0]	[1/eV]	[1/eV]	[1/eV]	[eV]	[meV]	[meV]	[g/cm ³]	[-]	[-]	
3.5	-	-	0.19	-	-	0.187	-	-	12.0	-	99.5	6.1	9.5	5.35	[1] 1975
3.5	5.00	-	0.19	1.00	-	0.187	-	-	12.0	-	92.0	6.1	9.5	5.35	[2] 1993
3.5	5.00	-	0.19	0.7	-	0.187	-	-	12.0	-	92.0	6.1	9.5	5.35	[3] 1995
3.4	5.50	5.60	0.19	0.41	0.41	-	-	-	10.1	-	92.0	6.095	9.5	5.35	[4] 1995
3.5	5.50	5.60	0.20	0.40	0.60	0.183	0.065	0.029	8.3	92.9	92.9	6.1	8.9	5.35	[5] 1997
3.39	5.39	5.59	0.20	0.40	0.60	0.189	0.067	0.029	8.3	91.2	91.2	6.15	8.9	5.35	[6] 1998
3.5	4.99	5.25	0.20	0.24	0.40	0.19	0.17	0	7.8	65.0	92.0	6.095	9.5	5.35	[7] 1998
3.5	5.40	5.60	0.20	1.00	1.00	0.189	0	0	8.3	91.2	91.2	6.15	8.9	5.35	[8] 1999
3.5	5.50	5.60	0.19	0.40	0.60	0.183	0.065	0.029	10.1	92.0	92.0	6.1	8.9	5.35	[9] 1999
3.5	5.45	5.60	0.21	0.25	0.40	0.19	0.1	0	8.0	65.0	92.0	6.095	9.5	5.35	[10] 2000
3.36	-	-	0.20	-	-	-	-	-	10.1	-	92.0	6.095	9.5	5.35	[11] 2000
3.52	5.77	5.87	0.212	-	-	0.37	-	-	8.3	65.8	90.88	6.087	9.7	5.28	[12] 2001
3.5	4.5	4.6	0.186	0.40	0.60	0.189	0.065	0.029	8.3	-	99.5	6.15	9.5	5.35	[13] 2002
3.52	5.77	5.87	0.212	0.493	0.412	-	-	-	8.3	-	90.88	6.087	9.7	5.28	[14] 2002
3.5	5.60	3.90	0.20	0.60	0.22	0.183	0.029	0.065	8.3	80.0	26&92.2	6.15	9.95	5.35	[15] 2004
3.39	5.29	5.59	0.20	1.00	1.00	0.189	0	0	8.3	92.0	92.0	6.15	8.9	5.35	[16] 2005
3.39	5.29	5.49	0.20	0.30	0.40	0.189	0	0	8.3	91.0	92.0	6.1	8.9	5.35	This work

Table 2: Summary of material parameters of wurtzite AlN for Monte Carlo simulation.

Bandgap energy			Electron mass			Non-parabolicity			Scattering models					Reference	
Γ_1	U	Γ_3	m_{Γ_1}	m_U	m_{Γ_3}	α_{Γ_1}	α_U	α_{Γ_3}	ADP	$\hbar\omega_{ij}$	$\hbar\omega_{LO}$	ρ	ε_s	ε_∞	Year
[eV]	[eV]	[eV]	[m_0]	[m_0]	[m_0]	[1/eV]	[1/eV]	[1/eV]	[eV]	[meV]	[meV]	[g/cm ²]	[-]	[-]	
6.20	6.90	-	0.48	1.0	-	0.044	0	-	9.5	99.2	99.2	3.23	8.5	4.77	[17] 1998
5.84	7.00	8.29	0.326	0.384	0.473	0.29	-	-	9.5	75.8	110.3	3.23	8.5	4.46	[12] 2001
6.00	7.05	8.49	0.26	0.495	0.55	0.207	0.035	0.023	-	76.1	110.7	-	-	4.68	[18] 2002
6.20	6.90	8.20	0.33	0.40	0.50	0.044	0	0	9.5	99.2	99.2	3.23	8.5	4.77	This work

model parameters for the acoustic deformation potential (ADP) scattering, inter-valley scattering ($\hbar\omega_{ij}$), and polar optical phonon scattering ($\hbar\omega_{LO}$). ε_∞ and ε_s are the optical and static dielectric constants, ρ is the mass density.

Our choice on the bandgap energies for GaN is based on one of the recent publications [16]. The particular setup for the masses has negligible impact within the available range, thus an average value similar to [10] has been chosen. Further parameters have been considered accordingly.

A complete set of material parameters for AlN is needed to model AlGaN, which is relevant for high-frequency amplifiers. We found only few publications which provide such parameter sets. For the simulation of AlGaN, an alloy scattering potential of 0.91 eV [18] is assumed. All other parameters are linearly interpolated between GaN and AlN.

3. Piezo-Scattering

An interesting result of the literature search performed for this work was the fact that in almost all MC simulations the piezo-scattering mechanisms were modeled assuming a cubic crystal structure. This is a correct approach to most of the technologically significant semiconductors, whereas for wurtzites the hexagonal structure has to be accounted for in the relevant piezo-scattering model.

In nitride crystals with wurtzite structure, elastic strain may be accompanied by macroscopic electric fields. This piezoelectric effect provides an additional coupling between the electron and acoustic vibrations [19].

The role of piezoelectric interaction in bulk wurtzite GaN has been recently analyzed by Kokolakis et al. [20]. In particular, the effect of acoustic piezoelectric scattering is taken in consideration, and the scattering rates have been calculated including the effect of screening. In accordance with their simulations, our results show that the piezo-acoustic rates are higher in the wurtzite phase than in the cubic phase, and they are very sensitive to the background doping of the sample. Since nitrides exhibit the largest piezoelectric constants among all of the III-V semiconductors an accurate modeling of piezoelectric scattering is especially important. In this work we present a piezoelectric scattering model similar to [19, 20], assuming equipartition, valid at temperatures over one Kelvin, considering non-parabolicity and screening in terms of the Thomas-Fermi inverse length q_0 .

According to the Fermi Golden rule the probability rate for scattering from a state \mathbf{k} to state \mathbf{k}' is:

$$W(\mathbf{k} \rightarrow \mathbf{k}') = W_a + W_e = \frac{2\pi}{\hbar} |F(\mathbf{q})|^2 [n_{\mathbf{q}} \delta(\epsilon(\mathbf{k} + \mathbf{q}) - \epsilon(\mathbf{k}) - \hbar\omega_{\mathbf{q}}) + (n_{\mathbf{q}} + 1) \delta(\epsilon(\mathbf{k} - \mathbf{q}) - \epsilon(\mathbf{k}) - \hbar\omega_{\mathbf{q}})]$$

where W_a corresponds to absorption ($\mathbf{k}' = \mathbf{k} + \mathbf{q}$) and W_e to emission ($\mathbf{k}' = \mathbf{k} - \mathbf{q}$) of a phonon with wave vector \mathbf{q} . The averaged coupling constant is obtained as:

$$|F(q)|^2 = C_0 K_{av} f(q); \quad f(q) = \frac{q^3}{(q^2 + q_0^2)^2}$$

The dimensionless constant K_{av} is used to introduce the mean sound velocity v_s which determines the energy of the averaged acoustic phonons $\hbar\omega_q = \hbar v_s q$. Employing the equipartition approximation $n_q = kT/\hbar\omega_q = kT/\hbar v_s q$, we obtain the scattering rates for emission and absorption.

$$\begin{aligned} \lambda_e &= C_1(\epsilon(k)) \int_0^{x_3} \frac{x^3}{(x^2 + 1)^2} dx - C_2(\epsilon(k)) \int_0^{x_3} \frac{x^4}{(x^2 + 1)^2} dx \\ \lambda_a &= C_1(\epsilon(k)) \int_{x_1}^{x_2} \frac{x^3}{(x^2 + 1)^2} dx + C_2(\epsilon(k)) \int_{x_1}^{x_2} \frac{x^4}{(x^2 + 1)^2} dx \end{aligned}$$

where $x = q/q_0$ is the normalized wave vector, and C_1, C_2 are sound velocity dependant coupling constants, proportional to K_{av} .

Almost all the works listed in Table 1 used a piezo-scattering model valid only for cubic crystals. In such a model the electromechanical coupling coefficient $K_{av,C}$ can be calculated by (1) with the piezo-coefficient e_{14} and the longitudinal and transverse elastic constants c_L and c_T [19]. The latter can be obtained from (2) and (3), respectively, from the electromechanical coupling coefficients c_{11}, c_{12} , and c_{44} or from the longitudinal and transverse sound velocities v_{sl} and v_{st} , if known.

$$K_{av,C}^2 = e_{14}^2 \cdot (12/c_L + 16/c_T) / (35 \cdot \epsilon_s) \quad (1)$$

$$c_L = 0.6 \cdot c_{11} + 0.4 \cdot c_{12} + 0.8 \cdot c_{44}; \quad v_{sl} = \sqrt{c_L/\rho} \quad (2)$$

$$c_T = 0.2 \cdot c_{11} - 0.2 \cdot c_{12} + 0.6 \cdot c_{44}; \quad v_{st} = \sqrt{c_T/\rho} \quad (3)$$

Table 3: Summary of elastic constants of GaN and AlN and the resulting longitudinal and transverse elastic constants and sound velocities.

c_{11} [GPa]	c_{12} [GPa]	c_{44} [GPa]	GaN Data Refs.	c_L [GPa]	c_T [GPa]	v_{sl} [m/s]	v_{st} [m/s]	c_{11} [GPa]	c_{12} [GPa]	c_{44} [GPa]	AlN Data Refs.	c_L [GPa]	c_T [GPa]	v_{sl} [m/s]	v_{st} [m/s]
296	120	24	exp. [21]	245	50	6342	2855	345	125	118	exp. [22]	351	115	10430	5962
374	106	101	exp. [23]	348	114	7557	4331	411	149	125	exp. [24]	406	127	11214	6280
390	145	105	exp. [25]	376	112	7859	4290	410	140	120	exp. [26]	398	126	11100	6246
377	160	81	exp. [27]	355	92	7637	3888	380	114	109	calc. [28]	361	119	10569	6060
365	135	109	exp. [29]	360	111	7693	4278	464	149	128	calc. [30]	440	140	11677	6579
370	145	90	exp. [26]	364	108	7733	4212	424	103	138	calc. [31]	406	147	11211	6746
373	141	94	exp. [32]	355	103	7641	4110	398	140	96	calc. [33]	372	109	10726	5814
369	94	118	calc. [31]	353	126	7620	4546	396	137	116	calc. [34]	385	121	10920	6131
396	144	91	calc. [33]	368	105	7775	4153	398	142	127	calc. [35]	397	127	11089	6280
367	135	95	calc. [34]	350	103	7585	4122								
350	140	101	calc. [35]	347	103	7548	4106								

Table 4: Summary of piezo coefficients of GaN and AlN for Monte Carlo simulation of piezo scattering.

e_{15} [C/m ²]	e_{31} [C/m ²]	e_{33} [C/m ²]	GaN Data Refs.	$\langle e_L^2 \rangle$ [C ² /m ⁴]	$\langle e_T^2 \rangle$ [C ² /m ⁴]	e_{15} [C/m ²]	e_{31} [C/m ²]	e_{33} [C/m ²]	AlN Data Refs.	$\langle e_L^2 \rangle$ [C ² /m ⁴]	$\langle e_T^2 \rangle$ [C ² /m ⁴]
-0.30	-0.36	1.00	exp. [36]	0.103	0.123	-0.48	-0.58	1.55	exp. [22]	0.251	0.304
-	-0.55	1.12	exp. [37]	0.175	0.234	-	-0.60	1.50	exp. [37]	0.260	0.334
-	-0.33	0.65	calc. [1]	0.061	0.082	-0.29	-0.58	1.39	exp. [38]	0.102	0.230
-	-0.49	0.73	calc. [39]	0.118	0.149	-	-0.60	1.46	calc. [39]	0.251	0.326
-0.22	-0.22	0.43	calc. [40]	0.027	0.036	-	-0.38	1.29	calc. [35]	0.169	0.187
-	-0.32	0.63	calc. [35]	0.058	0.077	-	-0.64	1.80	calc. [41]	0.349	0.429
-	-0.44	0.86	calc. [41]	0.109	0.145						

Table 3 summarizes the experimental and theoretical values of the elastic constants c_{11} , c_{12} , and c_{44} , available for wurtzite GaN and AlN in the literature. From them we calculated the corresponding c_L , c_T , v_{sl} , and v_{st} . We adopted the latest experimental values for GaN [32] and AlN [26] in our MC simulation. Note, that there are also experimental and theoretical values of the elastic constants for cubic GaN and AlN, but these are not relevant for this work.

Table 4 summarizes the experimental and theoretical values of the piezo coefficients e_{15} , e_{31} , and e_{33} , available for GaN and AlN in the literature. In cases, where e_{15} is not available $e_{15} = e_{31}$ is assumed. From them, we calculated the corresponding $\langle e_L^2 \rangle$ and $\langle e_T^2 \rangle$, which are necessary to obtain the coupling coefficient $K_{av,WZ}$ by (4) taking into account the wurtzite structure.

$$\begin{aligned}
 \langle e_L^2 \rangle &= 1/7 \cdot e_{33}^2 + 4/35 \cdot e_{33} (e_{31} + 2 \cdot e_{15}) + 8/105 \cdot (e_{31} + 2 \cdot e_{15})^2 \\
 \langle e_T^2 \rangle &= 2/35 \cdot (e_{33} - e_{31} - e_{15})^2 + 16/105 \cdot (e_{33} - e_{31} - e_{15}) + 16/35 \cdot e_{15}^2 \\
 K_{av,WZ}^2 &= (\langle e_L^2 \rangle / c_L + \langle e_T^2 \rangle / c_T) / \varepsilon_s
 \end{aligned} \tag{4}$$

In order to compare MC simulations which consider the piezo-scattering we took the extreme values of c_L and c_T from Table 3 and computed the value ranges for cubic $K_{av,C}$ using (1). Only in the case of [14] where c_L and c_T are provided, K_{av} can be estimated exactly. The results are presented in Table 5, where it is getting obvious that most authors either underestimate or even neglect the piezo-scattering effect, or use a higher e_{14} value in their MC simulations to arrive at hexagonal-like piezo-scattering rates using (1). On the other hand, most of the MC simulations assume relatively low sound velocity values which lead to an overestimation of the ADP scattering rate. This is another reason for many authors to find piezo-scattering to be negligible compared to ADP scattering, which we observe is not always the case.

Table 5: Parameter values for the piezo scattering model.

Ref.	v_{sl} [m/s]	v_{st} [m/s]	e_{14} [C/m ²]	c_L [GPa]	c_T [GPa]	K_{av} [-]
[1]	5000	-	0.560	245-376	50-126	0.13 - 0.20
[4]	4330	-	none	-	-	-
[5]	6600	2700	0.375	245-376	50-126	0.09 - 0.13
[6]	6560	2680	0.375	245-376	50-126	0.09 - 0.13
[7]	4330	-	0.560	245-376	50-126	0.13 - 0.20
[8]	6560	2680	0.375	245-376	50-126	0.09 - 0.13
[9]	4330	-	none	-	-	-
[10]	4330	-	none	-	-	-
[11]	6560	-	none	-	-	-
[12]	7619	-	0.375	245-376	50-126	0.09 - 0.13
[13]	4330	-	0.375	245-376	50-126	0.09 - 0.13
[14]	7619	-	0.368	266	62	0.117
[15]	8000	4860	0.375	245-376	50-126	0.09 - 0.13
[16]	6560	-	none	-	-	-
This work	7641	4110	-	355	103	0.137

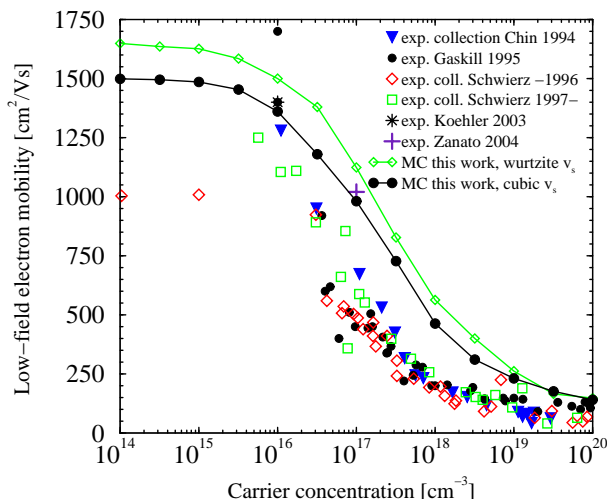


Figure 1: Low-field electron mobility as a function of carrier concentration in GaN. Comparison of the MC simulation results and experimental data.

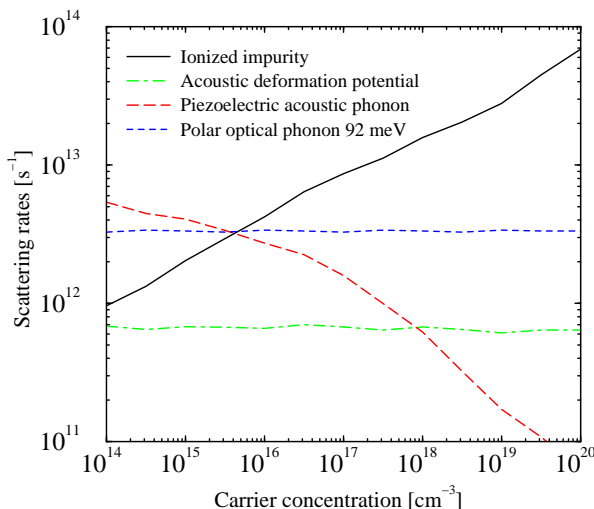


Figure 2: Illustration of the scattering rates in our simulation for wurtzite GaN as a function of doping concentration at 300 K.

4. Simulation Results

Using the established setup of models and model parameters, we obtained MC simulation results for different physical conditions (doping, temperature, field, etc.) for bulk GaN, AlN, and AlGaIn. Fig. 1 shows the low-field electron mobility in hexagonal GaN as a function of free carrier concentration. Two MC simulation curves are included to demonstrate the effect of the difference between the cubic and hexagonal sound velocities and their impact on the low-field mobility. The mobility depends on the sound velocity via the piezo and the ADP scattering mechanism. A higher sound velocity reduces the ADP scattering rate, which results in an increased mobility as illustrated in Fig. 1.

Fig. 2 shows the corresponding scattering rates as a function of the doping concentration in hexagonal GaN. Note, that the piezoelectric scattering is the dominant mobility limitation factor at low concentrations even at room temperature, beside the commonly accepted importance at low temperatures.

Our MC simulation is in fairly good agreement with experimental data from collections or single point measurements from [42, 43, 44, 45, 46]. The electron mobilities, selected for comparisons in this work, consider bulk material and are measured using the Hall effect. The discrepancy between our simulation results and the measured data might be attributed to dislocation scattering which is not considered in our work. This mechanism is considered to be a source of mobility degradation for GaN samples.

Numerous publications on GaN heterostructure devices (see e.g. a summary in [47]) provide inversion layer mobilities which are higher. These values are derived from transit frequency and device dimensions. However, two-dimensional electron gas heterostructures are plagued, among others, by surface scattering effects, and are not considered in this work.

Fig. 3 shows the low-field electron mobility as a function of lattice temperature in GaN at 10^{17} cm^{-3} concentration. The experimental data are from [46, 48, 49]. Note, that mobility increases over the years because of the improved material quality (reduced dislocation density).

Fig. 4 provides the electron drift velocity versus the electric field. We compare our MC result with other simulations [4, 5, 7, 12, 15, 50], and with the available experimental data [51, 52]. The low field data points are in qualitatively good agreement, at higher fields experimental values are significantly lower. Both experiments [51, 52] of electron velocities in bulk GaN, employ pulsed voltage sources. Many devices with etched constrictions were measured and the peak electron drift velocity $v_{d,max}$ was typically found to be about $2.5 \times 10^7 \text{ cm/s}$ at electric fields $E_{pk}=180 \text{ kV/cm}$. An overview of characteristic MC simulation results of GaN is summarized in Table 6. The discrepancy in the MC results comes from differently chosen sets of parameter values and considerations of scattering mechanisms.

A recent publication of Brazis et al. [15] on MC simulation of GaN provided a good fit to experimental results. It introduces a satellite valley close to the bottom of the conduction band and additional low-energy optical phonon

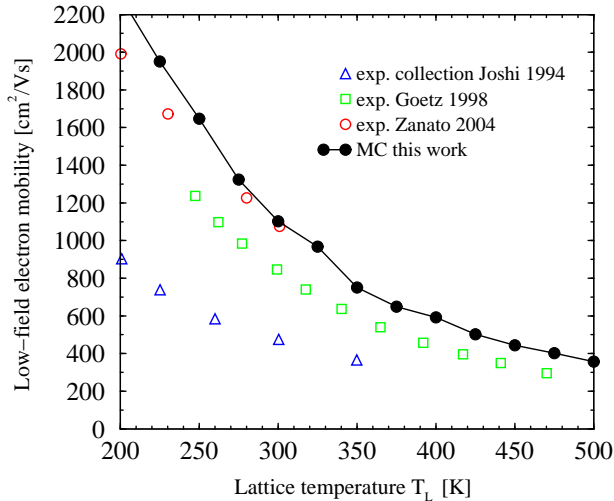


Figure 3: Low-field electron mobility as a function of lattice temperature in GaN at carrier concentration of 10^{17} cm^{-3} .

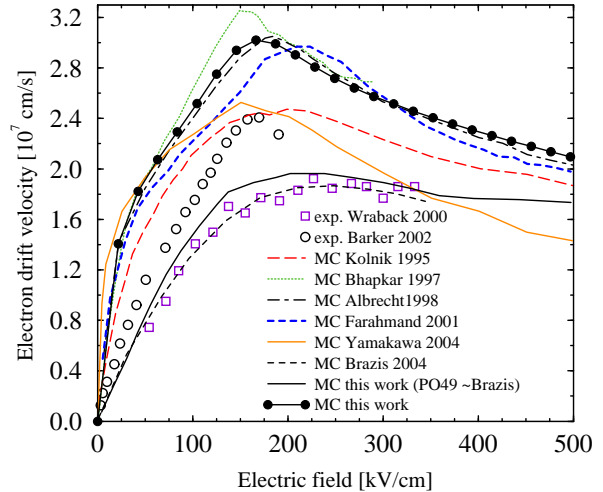


Figure 4: Drift velocity vs. electric field in wurzite GaN: Comparison of MC simulation results and experimental data.

Table 6: Comparison of drift velocity characteristic results of GaN MC.

Reference	[1]	[2]	[3]	[4]	[5]	[6]	[7]	[8]	[9]	[10]	[11]	[12]	[13]	[14]	[15]	[16]	[50]	This work
$v_{d,max} 10^5 [\text{m/s}]$	2.0	2.7	2.8	2.5	3.2	3.1	3.0	2.9	3.3	2.6	2.6	3.0	2.5	2.6	1.9	2.5	2.5	3.1
$E_{pk} [\text{kV/cm}]$	100	130	160	190	150	150	195	140	180	150	180	200	200	150	225	180	150	166
$v_{sat} 10^5 [\text{m/s}]$	-	1.5	1.6	1.9	2.5	2.5	2.1	1.4	1.5	1.8	2.0	1.9	1.7	2.4	1.8	1.3	1.4	2.1

modes of 26 meV. Using the MC parameter setup proposed by Brazis et al. and an averaged value for the polar optical phonon energy, between 92 meV and 26 meV, enabled us to arrive at a similar simulation result, validating the experimental result of [52] for a thin GaN film on a sapphire substrate.

Fig. 5 compares our MC simulation result for AlN against others from [12, 17, 53]. Our simulation results are in good agreement with [12, 17], since we use similar MC parameters as shown in Table 2. The difference visible at high fields can be explained by different effective electron masses used in the higher valleys. The simulation of [53] differs at low fields, since it ignores some mechanisms, e.g. ionized-impurity scattering.

Fig. 6 shows calculated electron steady-state drift velocity versus applied electric field in $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$. Our choice for a mole fraction of 0.2 in the illustration is based on the observation that alloy compositions in a range between 0.15 and 0.3 promise high mobilities [11, 12] and thus, e.g., highest transit frequencies in amplifier devices. As can be seen from Fig. 6 our result is in good agreement with other MC simulation results [11, 12]. Although an exact comparison cannot be performed, experimental data of Barker et al. [54] from $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}/\text{GaN}$ heterostructure are added.

5. Conclusion

Development of models for carrier transport in GaN-based materials gains importance. After careful review of available experimental and theoretical achievements, we developed a rigorous MC model, with the focal point on the piezo-scattering mechanisms in hexagonal crystal structures. It enables to understand the important effects taking place in this material system, and it allows creating analytical models for predictive device simulation.

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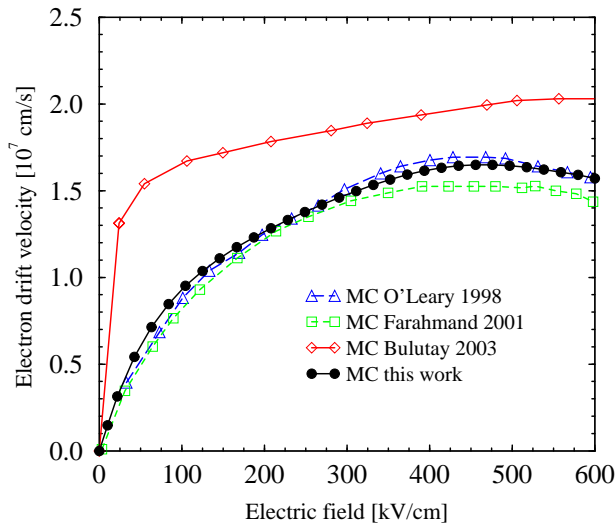


Figure 5: Drift velocity vs. electric field in wurzite AlN: Comparison to other MC simulations.

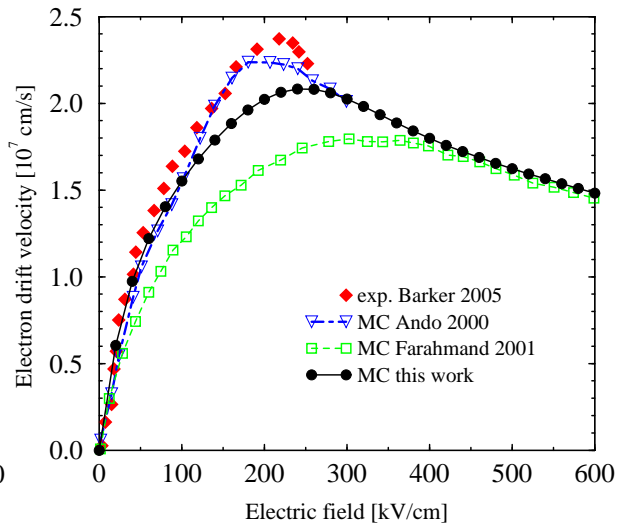


Figure 6: Drift velocity vs. electric field in wurzite Al_{0.2}Ga_{0.8}N: Comparison to other MC simulations.

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